## Acta Cryst. (1961). 14, 75

Some AB<sub>3</sub> compounds of the transition metals. By A. E. DWIGHT, J. W. DOWNEY and R. A. CONNER, JR., Argonne National Laboratory, Lemont, Illinois, U.S.A.

(Received 2 May 1960 and in revised form 15 July 1960)

## 1. Introduction

An investigation is underway to identify new intermetallic compounds of  $AB_3$  composition. This paper presents crystal structure data for fifteen  $AB_3$  compounds which have been studied since those reported in a paper by Dwight & Beck (1959).

## 2. Experimental methods

Alloys were made by arc melting 3- or 5-gram charges in a water-cooled copper crucible. All alloys were found to be sufficiently brittle that powder could be made by hammering the cast buttons in a closed carbide die. After screening through silk cloth, specimens of powder were wrapped in Mo foil, sealed in evacuated Vycor capsules and annealed. Diffraction patterns were taken on a Straumanis-type Debye–Scherrer camera of 114.6 mm. diameter. The Nelson–Riley extrapolation was used on patterns of the cubic  $Cu_3Au$ -type compounds to obtain

## Table 1. $AB_3$ compounds with the Cu<sub>3</sub>Au-structure, Ll<sub>2</sub>, space group Pm3m

| $AB_3$            | $a_0$   | $AB_3$   | $a_0$   | $AB_3$           | $a_0$   |
|-------------------|---------|----------|---------|------------------|---------|
| ScRh <sub>3</sub> | 3.900 Å | $YPt_3$  | 4∙075 Å | $ThRh_3$         | 4·139 Å |
| $ScPd_3$          | 3.981   | $LaPd_3$ | 4.235   | $\mathrm{URh}_3$ | 3·991   |
| $ScPt_3$          | 3.958   | $HoPd_3$ | 4.064   | UIr <sub>3</sub> | 4.023   |
| YPd.              | 4.074   | HoPt.    | 4.064   | -                |         |

## Table 2. $AB_3$ compounds with close packed ordered structures

| $AB_3$           | Type              | $c_0$   | $a_0$         | c/a   |
|------------------|-------------------|---------|---------------|-------|
| ThPd,            | TiNi <sub>2</sub> | 9·826 Å | 5·856 Å       | 1.678 |
| VPd,             | TiAl              | 7.753   | 3.847         | 2.015 |
| VPt <sub>2</sub> | TiAl              | 7.824   | <b>3</b> ·861 | 2.027 |
| HfAĬ.            | ZrAl              | 17.155  | 3.989         | 4.301 |

Table 3. Powder diffraction data for ThPd<sub>3</sub>, DO<sub>24</sub>, Cr  $K\alpha$  radiation, space group P6<sub>3</sub>/mmc

| hkl            | $d_o$   | $d_c$            | Io  | $I_c$      |
|----------------|---------|------------------|-----|------------|
| 102            | 3.506 Å | 3·523 Å          | w   | 30.9       |
| 110            | 2.910   | 2.927            | vvw | 17         |
| 200            | 2.526   | 2.535            | vw  | 15.9       |
| 004            | 2.444   | $2 \cdot 450$    | vs  | 78.5       |
| 202            | 2.247   | 2.252            | vs  | <b>200</b> |
| 203            | 1.997   | 2.003            | 8   | 46.4       |
| 114            | 1.876   | 1.879            | w   | 10         |
| 212            | 1.782   | 1.785            | vw  | 9.9        |
| 204            | 1.759   | 1.762            | w   | 10.7       |
| 205            | 1.5485  | 1.5506           | ms  | 24.8       |
| 220            | 1.4624  | 1.4636           | 8   | 62.9       |
| 304            | 1.3903  | 1.3912           | vw  | 6.7        |
| 206            | 1.3722  | 1.3730           | 8   | 72.3       |
| 312            | 1.3509  | 1.3516           | vw  | 7.8        |
| $\{224\\401\}$ | 1.2566  | 1.2565           | vs  | 192        |
| 401 J<br>402   | 1.2272  | 1.2571<br>1.2271 | 8   | 104.5      |
| $207 \\ 008 $  | 1.2251  | 1.2255           | 8   | 66•6       |
| 403 J          | 1.1819  | 1.1817           | ms  | 51.5       |

| Table 4. | Powder                       | diffraction | data for | • VPd | $l_3$ and | VPt <sub>3</sub> , | DO22, |
|----------|------------------------------|-------------|----------|-------|-----------|--------------------|-------|
|          | $\operatorname{Cr} K \alpha$ | radiation,  | space gr | roup  | 14/mm     | im -               |       |

|               |                  | $VPd_3$       |                              |                  | ${\rm VPt}_{\bf 3}$       |               | VNi <sub>3</sub> * |
|---------------|------------------|---------------|------------------------------|------------------|---------------------------|---------------|--------------------|
| hkl           | $\overline{I_o}$ | $d_o$ (Å)     | $d_c$ (Å)                    | $\overline{I_o}$ | <i>d</i> <sub>o</sub> (Å) | $d_c$ (Å)     | Io                 |
| 002           |                  |               |                              | vvw              | <b>3</b> ·86              | 3.912         | vvw                |
| 101           | w                | 3.427         | 3.446                        |                  |                           |               | w                  |
| 110           | vvw              | 2.70          | 2.720                        | vw               | 2.711                     | 2.730         | vvw                |
| 112           | vs               | $2 \cdot 219$ | 2.227                        | vs               | $2 \cdot 228$             | $2 \cdot 239$ | vs                 |
| 103           | vw               | $2 \cdot 142$ | $2 \cdot 145$                |                  | _                         |               | vvw                |
| 004           | m                | 1.933         | 1.938                        | m                | 1.949                     | 1.956         | m                  |
| 200           | ms               | 1.919         | 1.923                        | 8                | 1.923                     | 1.930         | ms                 |
| 202           | vw               | 1.719         | 1.723                        | w                | 1.726                     | 1.731         | vvw                |
| 211           | w                | 1.678         | 1.680                        | w                | 1.703                     | 1.686         | vw                 |
| 114           | vvw              | 1.576         | 1.579                        | w                | 1.588                     | 1.590         | vvw                |
| $105 \\ 213 $ | w                | 1.430         | $1 \cdot 438 \\ 1 \cdot 432$ |                  |                           |               | vvw<br>vw          |
| 204 Ó         | 8                | 1.364         | 1.365                        | 8                | 1.3724                    | 1.3740        | 8                  |
| 220           | ms               | 1.359         | 1.360                        | ms               | 1.3636                    | 1.3649        | ms                 |
| 222           | w                | 1.284         | 1.283                        | w                | 1.2879                    | 1.2888        | vvw                |
| 301           | w                | 1.265         | 1.265                        | vw               |                           |               | vvw                |
| 310           | w                | 1.2163        | 1.2165                       | w                | 1.2205                    | 1.2208        |                    |
| 116           | 8                | 1.1672        | 1.1672                       | vs               | 1.1765                    | 1.1767        |                    |
| 312           | vs               | 1.1606        | 1.1606                       | vvs              | 1.1653                    | 1.1654        |                    |
| 215           | m                | 1.1518        | 1.1518                       |                  |                           |               |                    |

\* VNi<sub>3</sub> data from Pearson & Hume-Rothery (1952).

## Table 5. Powder diffraction data for DO<sub>23</sub>-type compounds

|            |                  | $\frac{\mathrm{HfAl}_{3}}{\mathrm{Cr}\ K\alpha}$ | $\operatorname{ZrAl}_{3}^{*}$<br>Cu $K\alpha$ |
|------------|------------------|--|---|
| hkl        | $\overline{I_o}$ | $d_o$  | Io  |
| 004        | m                | 4.25   | <i>s–m</i>                                    |
| 101        | ms               | 3.86   | s-m   |
| 103        | m                | 3.25   | 8   |
| 110        | m                | 2.81   | 8   |
| 105        | ms               | 2.592  | s-m   |
| 114        | vs               | 2.351  | st  |
| 008        | m                | 2.143  | m-st  |
| 200        | ms               | 1.993  | st  |
| 204        | m                | 1.810  | 8   |
| 211        | m                | 1.774  | 8   |
| 109;00,10  | m                | 1.722  | 8   |
| 213        | m                | 1.707  | 8   |
| 215        | m                | 1.581  | 8   |
| 208        | 8                | 1.460  | m-st  |
| 217        | w                | 1.442  | 888   |
| 220        | ms               | 1.410  | s-m   |
| <b>224</b> | m                | 1.339  | 888   |
| 301        | w                | 1.325  | <i>888</i>                                    |
| 219;20,10  | ms               | 1.302  | 8   |
| 11, 12     | 8                | 1.275  | m   |
| 310        | m                | 1.261  | 88  |
| 10, 13     | m                | 1.253  | 88  |
| 305        | m                | 1.239  | 888   |
| 314        | vs               | $1 \cdot 2096$                                   | m-st  |
| 228        | vs               | 1.1780   | m   |
| 307        | vw               | 1.1683   |   |
| 20, 12     | ms               | 1.1619   | 888   |
| 316        | ms               | 1.1540   |   |

\* ZrAl<sub>3</sub> intensities taken from Brauer (1934).

a more accurate lattice parameter. Cohen's least-squares method was applied to the hexagonal and tetragonal patterns.

## 3. Experimental data

The compounds and lattice parameters found in this investigation are listed in Tables 1 and 2. The observed and calculated intensities and d spacings for ThPd<sub>3</sub> of the hexagonal TiNi<sub>3</sub>-type are given in Table 3. Table 4 lists the observed and calculated d spacings and observed intensities for two compounds, VPd<sub>3</sub> and VPt<sub>3</sub>, of the TiAl<sub>3</sub>-type. No intensity calculations were made for VPd<sub>3</sub> and VPt<sub>3</sub>, but the observed intensities for these compounds are in good agreement with those published by Pearson & Hume-Rothery (1952) for the VNi<sub>3</sub> compound. The intensities observed by Pearson & Hume-Rothery for VNi<sub>3</sub> are reproduced in Table 4.

Table 5 lists similar data for the HfAl<sub>3</sub> compound, which is isostructural with ZrAl<sub>3</sub>. No intensity calculations were made for HfAl<sub>3</sub>, but the observed intensities for HfAl<sub>3</sub> are in good agreement with Brauer's data for ZrAl<sub>3</sub>, considering that the HfAl<sub>3</sub> intensities were obtained with Cr  $K\alpha$  radiation, and the ZrAl<sub>3</sub> intensities with Cu  $K\alpha$ .

## 4. Discussion

With the discovery that  $ThPd_3$  has the  $TiNi_3$  structure, it is noted that the sequence of compounds  $TiPd_3$ ,  $ZrPd_3$ ,  $HfPd_3$ ,  $ThPd_3$  and  $UPd_3$  are isostructural. Likewise, the series  $TiRh_3$ ,  $ZrRh_3$ ,  $HfRh_3$ ,  $ThRh_3$  and  $URh_3$  all have the Cu<sub>3</sub>Au structure. The two compounds  $VPd_3$  and  $VPt_3$  are isostructural with  $VNi_3$ .  $VPd_3$  was previously reported by Koster & Haehl (1958).

The TiAl<sub>3</sub> family of compounds, of which VPd<sub>3</sub> and VPt<sub>3</sub> are members, has been indexed in the older literature with respect to two different unit cells. In this investigation we use the smaller unit cell proposed by Pearson & Hume-Rothery (1952) for VNi<sub>3</sub>. The  $c_0$  parameters are the same for the two unit cells; the  $a_0$  parameters are related by the factor  $1/\sqrt{2}$ .

The compound  $HfAl_3$  is closely related to the  $TiAl_3$ type compounds, as was shown by Brauer (1934). The structure of  $HfAl_3$  may be envisioned as containing two  $TiAl_3$  unit cells stacked end to end with one inverted, and with certain layers rippled in the  $HfAl_3$  structure as opposed to flat layers in the  $TiAl_3$  structure.

The authors wish to acknowledge helpful discussions with Dr M. V. Nevitt. This work was performed under the auspices of the U.S. Atomic Energy Commission.

#### References

BRAUER, G. (1934). Z. Anorg. Chem. 242, 1.

DWIGHT, A. E. & BECK, P. A. (1959). Trans. Amer. Inst. Min. (Metall.) Engrs. 215, 976.

KOSTER, W. & HAEHL, W. (1958). Z. Metallk. 49, 647.

PEARSON, W. B. & HUME-ROTHERY, W. (1952). J. Inst. Met. 80, 641.

## Acta Cryst. (1961). 14, 76

# Structure cristalline du chloro 9-bromo 10-anthracène par diffraction des rayons X. Par M. HOSPITAL, Laboratoire de Minéralogie et de Rayons X, Faculté des Sciences, Université de Bordeaux, France

(Reçu le 15 juillet 1960)

Le chloro 9-bromo 10-anthracène est isomorphe du dibromo 9-10 anthracène (Trotter, 1958). Les mailles tricliniques sont très semblables.

|          | Dibromo 9-10 anthracène<br>(Trotter) | Chloro 9-bromo<br>10-anthracène |
|----------|--------------------------------------|---------------------------------|
| a        | $8,88 \pm 0.02$ Å                    | $8,87 \pm 0,02$ Å               |
| ь        | $16,15 \pm 0,04$                     | $16,14 \pm 0,03$                |
| с        | $4,06 \pm 0,01$                      | $4,06 \pm 0,01$                 |
| $\alpha$ | $97^{\circ} \ 05' \pm 10'$           | $97^{\circ} 10' \pm 10'$        |
| β        | $100^{\circ} 21' \pm 10'$            | $100^{\circ} \ 39' \pm 10'$     |
| γ        | $98^{\circ} 50' \pm 10'$             | $98^{\circ} \ 27' \pm 10'$      |
| v        | 559,3 Å <sup>3</sup>                 | 558,2 Å <sup>3</sup>            |

Densité mesurée: 1,75 g.cm.<sup>-3</sup>. Densité calculée: 1,735 g.cm.<sup>-3</sup>.

Les noyaux anthracéniques occupent exactement les mêmes positions dans les 2 structures. Les molécules présentent un centre de symétrie comme dans le dibromo 9-10 anthracène.

La structure du chloro 9-bromo 10-anthracène est une

structure statistique analogue à celle du chloro 1-bromo 4-benzène (Klug, 1947). Les atomes lourds sont des atomes mixtes:  $\frac{1}{2}(Br + Cl)$ .

La distance  $C_9$ -atome lourd est de  $1,88 \pm 0,01$  Å, intermédiaire entre  $C_9-Cl = 1,74$  Å dans le dichloro 9-10 anthracène (Trotter, 1959) et  $C_9-Br = 1,93$  Å du dibromo 9-10 anthracène (Trotter, 1958). Les atomes de carbone  $C_9$  et  $C_{10}$  du noyau anthracénique sont cependant bien localisés et ne sont pas déplacés vers le chlore comme semble l'exiger la différence de 0,2 Å entre la liaison C-Br et la liaison C-Cl.

Les déformations du noyau anthracénique que l'on peut constater dans la structure du dibromo 9-10 anthracène (Trotter, 1958) se conservent intégralement dans la structure du chloro 9-bromo 10-anthracène.

## Références

KLUG, A. (1947). Nature, Lond. 160, 570. TROTTER, J. (1958). Acta Cryst. 11, 803. TROTTER, J. (1959). Acta Cryst. 1, 54.